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Polymer opals as novel photonic materials

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Abstract

Synthetic opals, based on self-assembled arrays of core-shell (bead-matrix) polymer microparticles, are a promising platform for next-generation bulk-scale photonic structures, coatings, fibres and sensors. This perspective article highlights recent work in this area, ranging from characterization and application, to advances in more fundamental understanding of structural colour effects. These advances include viscoelastically tuned symmetry breaking, the observation of anisotropic optical scattering and the study of polymer opals as an analogue to the intrinsically disordered, low-refractive-index contrast systems associated with opals seen in nature.

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INTRODUCTION

The novel use of soft nanomaterials in the design of photonic structures, with macroscale internal structural bulk ordering, presents opportunities for a step-change away from the monolithic architectures that are currently relied upon. Photonic crystals, where periodic variations in refractive index create photonic 'band gaps', allow the development of structured materials with distinguishing optical properties (e.g. structural colour), which are not accessible using dyes or pigments.

In recent work, a low-cost, industrial-scale technique to produce flexible opals has been developed using melting and shearordering of core – shell polymer nanoparticles. 1,2 This is in marked contrast to many previous strategies for constructing photonic crystals, which have relied upon top-down lithography³ or solvent-driven self-assembly of high- and low-refractive-index components.⁴ Even simple structures, such as photonic crystal face-centred cubic (fcc) lattices, based on these methods cannot be fabricated in any scalable fashion, and the resultant structures notably lack any mechanical robustness. No precedents exist for the application of shear-ordering techniques to these granular solvent-free systems, which allow formation of permanent, mechanically robust composites in the solid state (Fig. 1). The resultant structures are low-defect flexible polymer fcc opal films, with fundamental optical resonances tunable across the visible and near-infrared regions (by varying the precursor particle size from 200 to 350 nm, and hence the resulting lattice parameter). In the lower refractive index contrast regime associated with these polymer composites ($\Delta n \approx 0.1$), colour generation arises through spectrally resonant scattering inside a three-dimensional (3D) fcc-lattice photonic crystal generating a 'scattering cone', 5,6 as opposed to normal reflective iridescence based on Bragg diffraction. In addition, one of the most attractive features of elastomeric polymer opals is the tunability of their perceived colour by the bending or stretching modification of the (111) plane spacing.

The fabrication technique employed uses shear-flow selfassembly of core-interlayer-shell polymer particles into 3D cubic lattices through extrusion at high temperatures. As illustrated in Fig. 1(b), the core-shell particle precursors are typically 200-300

nm in diameter, and consist of a hard polystyrene core, coated with a thin poly(methyl methacrylate) 'grafting' layer, and a soft poly(ethyl acrylate) outer shell. These particles are prepared using a multi-stage emulsion polymerization process. The material may be processed using a twin-screw mini-extruder and adjustable temperatures of 25–250 °C. Bulk quantities of the opal precursor materials (as shown in Fig. 1(a)) are manually driven into the extruder, where they form a melt and are homogenized under the extreme shear forces provided by recycling through the screws. The overpressure generated then drives the shear-ordered granular material through a narrow-bore stainless steel die, producing bulk opaline ribbons. In such polymer opal films, ordering proceeds from the outer flat surfaces which seed accumulation of (111) planes,⁷ but recent work has now shown that it is also possible to produce opaline fibres by extrusion in a wire geometry.8

Additionally, the introduction of a small (ca 0.05% by weight) fraction of carbon nanoparticle pigment into the interstices of the photonic crystal lattice does not disrupt the lattice quality, but results in a marked increase in the colour saturation of the opals with the concentration of carbon, as shown in Fig. 1(e).⁵ This principle is of fundamental interest in understanding the origins of structural colours and iridescence in natural opals, such as those in minerals or in biological structures. However, the main difficulty of this fabrication approach has been the lack of true bulk order.

A significant advance is achieved with an edge-induced rotational shearing (EIRS) process, which produces reproducible highly uniform samples with bulk ordering of sub-micrometre components, greatly enhancing both the intensity and chromaticity of the observed structural colour. 10 A schematic

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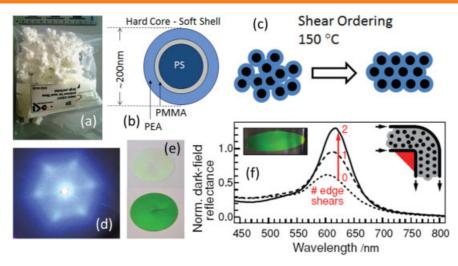


Figure 1. (a) Bulk batch of as-synthesized core–interlayer–shell (CIS) particles, used for polymer opal production. (b) Schematic of the CIS system, based on polystyrene (PS)–poly(methyl methacrylate) (PMMA)–poly(ethyl acrylate) (PEA). (c) Schematic of the fabrication technique using shear-flow self-assembly of polymer particles into 3D cubic lattices at high temperatures. (d) Transmission diffraction pattern taken at normal incidence for a *ca* 40 μm thick film of an edge-sheared opal with sphere diameter of 630 nm, the laser wavelength used was 405 nm. (e) Green polymer opal films showing the large enhancement of structural colour saturation with the addition of 0.05% by weight of carbon nanoparticles. (f) Dark-field spectra of polymer opal films show improving order with edge shearing. Data are normalized to a reference Lambertian scatterer. The insets show a photo of an edge-sheared polymer opal film (*ca* 6 cm × 2 cm, left) and edge-induced rotational shear processing of core–shell nanospheres between rigid poly(ethylene terephthalate) tapes (right). (Images in (e) and (f) reproduced with kind permission of the Optical Society of America⁵ and Wiley.²¹)

of this mechanically controlled process and the associated optical properties are shown in Fig. 1(f). In brief, the process consists of a combination of bulk extrusion, followed by rolling (linear shear) into ca 100 µm thick films and finally EIRS (across a polished metallic hot-edge, with an apex angle of $\alpha = 90^{\circ}$ and radius of curvature $<10 \mu m$) yielding permanent rolls of opaline film. As a powerful demonstration of the 3D bulk ordering achieved in these opaline films, optical diffraction experiments readily revealed very clear and distinctive diffraction-spot patterns, with a characteristic six-fold symmetry (Fig. 1(d)), which was precisely aligned to the processing direction in the EIRS process. Whilst the appearance of such well-resolved diffraction spots in transmission for the EIRS sample represents compelling evidence of crystallinity throughout the film of many hundred sphere layers, the appearance of six spots is also indicative of the presence of twinning and/or stacking faults within the cubic crystal structure.¹¹

APPLICATIONS OF POLYMER OPALS

The demonstration of reproducible scale-up of these elastomeric synthetic opaline films to industrial length scales makes them very attractive as a route to a wide range of large-area photonics applications, including sensors and coatings. In this perspective article, we aim to summarize some of the most notable and promising examples, within the context of the underlying science.

Figure 2(a) shows a representative image of polymer opals with strong and vivid iridescence, as defined by the core spacing and viewing angle. In addition to their subjectively decorative/aesthetic qualities, the unique structural colour properties of polymeric opals, together with intrinsic processability, stretchability and durability also potentially make them attractive for applications such as security or anti-forgery labelling in banknotes, credit cards and designer merchandise, for example. An illustration of how these favourable properties relate to applications is shown in Fig. 2(b), where a high-quality polymer opal film has been prepared as a novel textile, by attachment

onto a fabric backing, using the simple means of applying a cool iron-press. The sample shows all of the attractive properties of elastic stretch-, bend- and twist-tunability, in addition to a vivid structural colour effect.¹²

As an extension of the polymer processing technologies reviewed here, extruding high-quality opaline fibres in an industrially scalable fashion is now also practical (Fig. 2(c)).¹³ As with the opaline thin films, these fibres exhibit structural colour based on the self-assembly of sub-micrometre core – shell particles, with a spectrum which is stretch-tunable across the visible region. Fibres are directly extruded using modified circular dies of diameters ranging from 100 to 2000 µm. Since ordering during standard extrusion proceeds from the outer flat surfaces which seed accumulation of (111) planes, it was far from obvious that opal fibres could be produced. However, in addition to experimental demonstration, a dynamic theoretical (granular) model predicts the formation of ordered close-packed layers on the outer region of the fibre, and the growth of this order inwards from these seed layers. In both experiment and theory, fibres are characterized by an internal substructure, in which a concentric zone near the exposed surface develops particularly strong structural colour. After suitable UV-A photocrosslinking, the final fibre products have sufficient mechanical robustness to allow them to be handknitted into stretchable fabrics. These elastically tuned fibres are thus potential candidates for a novel range of nanomaterials and clothing fabrics, utilizing strong structural colour effects as a replacement for toxic and photodegradable dyes.

Large-scale shear-ordered photonic crystals have also been shown to exhibit unusual *thermochromic* properties.¹⁴ By balancing the refractive index of the polymer core and composite shell components at room temperature, transparent films are created, which become coloured on heating (Fig. 2(d)). Since this scattering-based structural colour depends only on resonant Bragg scattering of non-pigmented components, it can be tuned to any wavelength. Whilst it is expected that there is a thermally induced change in refractive index contrast, the observed colour



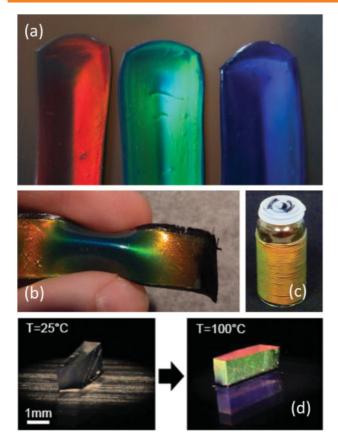


Figure 2. (a) Illustration of the striking iridescence of red, green and blue polymer opal thin films. Core particle sizes range from *ca* 300 nm (red) to *ca* 200 nm (blue). (b) A polymer opal mounted onto a Lycra fabric, illustrating the stretch-, bend- and twist-tunability. (c) Post-extruded long fibres of opaline quality. (d) Change in an iso-refractive opal structure as the temperature is raised, illustrating a clear thermochromic effect. (Images in (c) and (d) reproduced with kind permission of the Optical Society of America⁸ and the American Institute of Physics. ¹⁴)

shifts with temperature are not accounted for by simple theory, and probably reflect a more complex interaction between the core and shell refractive indices, and the mechanical expansion and strain effects upon heating. This thermochromic mechanism can also be harnessed in a wide variety of ways by tuning core and shell sizes and polymer compositions, pointing towards a potential use of such opals in thermal sensors and indicators, where large areas may be important (e.g. food packaging). Further optimization and incorporation of other (e.g. emissive) nanoparticles would also open a range of potential applications in displays and in switchable structural colour materials. ¹⁵

The prospects for utilizing such photonic materials in optoelectronic applications, such as photovoltaics or electrically tunable colour films, require conductive colloidal crystals which have been unavailable thus far. In the recent work of Imai et al., ¹⁶ electrically conductive polymeric 3D photonic crystals were prepared by the shear ordering of hetero-coagulated composites consisting of monodisperse core–shell polymer spheres and single-walled carbon nanotubes (SWNTs). The retention of strong iridescent colour indicates that the highly ordered polymer opal structures are not disrupted by the presence of the conductive nanotube networks. Thermal annealing leads to a significant increase in the overall electrical conductivity of thin-film samples yielding DC conductivities of 10⁻⁴ 5 cm⁻¹, with percolation

thresholds of less than 0.4 wt% of SWNTs. As well as this work being an important milestone in the quest for 'conductive photonics', it is anticipated that further optimization of processing parameters, such as those of sample annealing, will enable yet higher conductivities to be reached, making further practical applications feasible. Following on from the work of Imai *et al.*, electrically tuned photonic crystals have also been demonstrated by applying fields across shear-assembled elastomeric polymer opal thin films.¹⁷ With increasing applied electric fields, the polymer opal films stretch biaxially under Maxwell stress, deforming the nanostructure and producing marked colour changes. This electrooptic tuning of the photonic band gap is rapid (switching at <100 ms), and repeatable over many cycles, bridging the gap between electro-active materials and photonic crystals.

A PLATFORM FOR EXPLORING THE FUNDAMENTAL PHYSICS OF PHOTONIC CRYSTALS AND STRUCTURAL COLOUR

Besides the technological applications of polymer opals, such systems are interesting analogues of structural colour in nature, as well as the intensity of scatter from clouds and radar imaging. Additionally, the durable viscoelastic properties of the samples allow access to a whole host of experiments, and the associated scientific insights, which are mechanically impossible with more conventional monolithic photonic crystals.

Symmetry is a crucial ingredient in the self-assembly of nanostructures with novel optical, electronic, magnetic or thermal functionality. However, many new physical properties that are desirable to access depend on breaking this symmetry to introduce higher-order complexity, which has proved difficult.¹⁸ Polymer opals have been used as a paradigm to demonstrate elastically induced phase transitions to break the structural symmetry of self-assembled nanostructures, producing significantly modified functional properties. Stretching ordered polymer opals in different directions transforms the fcc photonic crystal into correspondingly distorted monoclinic lattices. This breaks the conventional selection rules for scattering from the crystal planes, yielding extra multiply scattered colours when the phasebreaking stretch is in specific directions. Scattering has been spectroscopically tracked in real time as the samples distort, revealing a new phase transition into a lower symmetry monoclinic lattice structure. 19 New, normally forbidden Bragg peaks become immediately visible (as the condition for fcc that Miller indices h, k, I must be all even or all odd is broken); such effects have not even been experimentally observed in atomic crystals.

Photonic properties have also been widely studied in natural biomaterials which self-assemble in more sophisticated ways,²⁰ such as butterfly wings, beetle carapaces and flower petals. However, such self-assembly processes intrinsically incorporate disorder in the nanostructures, which degrades the reflectivity but enhances the wider scattering of light (i.e. into 'scattering cones'). In that sense, polymer opals represent an important analogue by which a full understanding of the role of such disorder in the optical properties of such systems may be characterized. Recently, hyperspectral goniometry has been used with a polymer opal sample to demonstrate how to completely reconstruct the reciprocal space of scattering vectors.²¹ This yields the 3D shape of the dominant reciprocal lattice point responsible for the intense structural colour, and separates resonant and background scattering processes. In these crystals, a previously unknown but



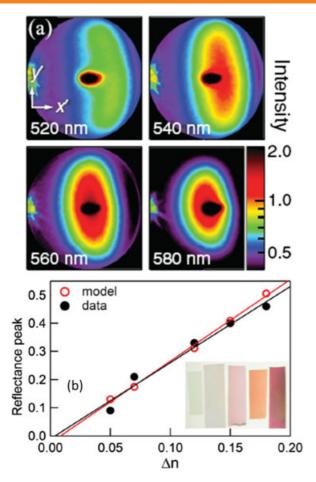


Figure 3. (a) Intensity pattern of scattered light from a polymer opal at various wavelengths, as projected onto a hemispherical screen, showing highly anisotropic scattering (images use a high dynamic range false-colour scale as indicated). The small black spot in the centre is the intense specular reflection; the hole through which white light is introduced is seen on the left. (b) Normal incidence reflectance peak heights plotted as a function of refractive index contrast, Δn , with linear fits to both experimental data and a one-dimensional multilayer quasi-model. In the model, layers had an inbuilt disorder of 7% of the interlayer thickness. The inset shows the samples (widths of ca 3cm) as viewed in transmitted white light, with the refractive index contrast increasing from left to right. (Images reproduced with kind permission of Wiley²¹ and the American Institute of Physics.²³)

significant anisotropy of the scattering process which dominates the optical interactions was identified. Such measurements yield a composite image at each incident wavelength (Fig. 3(a)) which captures the front surface specular reflection peak while clearly resolving the resonant scattering cone (which is up to two orders of magnitude less intense). Analysis reveals that the scattering is anisotropic in all three directions, appearing as a flattened ellipsoid. While an anisotropic response to stress has been previously reported (described above), this optical anisotropy is intrinsic to the nanostructure, appearing in all unstretched samples. This anisotropy has its origins in the directionalized ordering of the EIRS process and it was found that short chain defects govern the observed angular scattering cone. This implies that defects spatially localize light over length scales of many sphere periods, and their control thus plays a crucial role in self-assembly of structurally coloured photonic crystals.

The refractive index contrast (Δn) between the core beads and the matrix polymer is essential for the appearance of such

structural colour; however, it is not clear how the strength of such resonances changes as Δn increases. The refractive index contrast between core and shell materials can be varied by controlling the chemical composition of the shell material using low-refractiveindex fluorinated monomers (the fluorinated monomer used in the shell layer was 2,2,2-trifluoroethyl acrylate; Fig. 3(b), inset).²² The effect of Δn on the resonant scattering and reflectivity of polymer opals, within a regime of $\Delta n < 0.2$, has been recently studied. The strength of resonant Bragg reflection from polymer opals is found to vary linearly with Δn , in marked contrast to the expected quadratic build-up of Fresnel reflections scaling as $(\Delta n)^2$. This occurs due to the interplay of disorder and periodicity, in close agreement with a simple one-dimensional periodic model with inbuilt disorder, as shown in Fig. 3(b).²³ Hence, the degree of disorder in the sample demonstrably plays a fundamental role in these linear dependencies. In previous studies of polymer opals, we presented crystallographic evidence of stacking faults and/or twinning within the cubic structures, which introduce a certain level of disorder throughout the films. It is likely that these characteristics are also intrinsic to many self-assembled periodic structures, including those found in nature. 9,20,24

SUMMARY

Elastomeric polymer opals, in both thin-film and fibre form, can now be manufactured using ubiquitous polymer processing techniques and have many attractive functional features: intense structural colour, with inherent stretch- and bend-tunability, and excellent durability and mechanical robustness. As demonstrated, these opals are thus potential candidates for a novel range of nanomaterials, coatings, sensors and also clothing fabrics, utilizing structural colour effects as a replacement for toxic and photodegradable dyes.

In addition to these innovative applications, polymer opals have also proved to be a highly fruitful platform for more fundamental experimental studies of the physics of photonic crystals, of structural colour and of the self-assembled optical structures seen in nature. In particular, practical demonstrations of symmetry breaking in photonic crystals, and a deeper understanding of the roles of disorder, anisotropy and refractive index contrast in structural colour, have been achieved. We anticipate further developments in this regard in the future, and the extension of the archetypal core—shell polymer opal system into a yet wider range of engineered functionalities and applications, including novel biomimetic strategies.

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REFERENCES

- 1 Ruhl T, Spahn R and Hellmann GP, Polymer 44:7625 7634 (2003).
- 2 Pursiainen OLJ, Baumberg JJ, Winkler H, Viel B and Ruhl T, Appl Phys Lett 87:101902 (2005).
- 3 de la Rue R. Nature Mater **2**:74–76 (2003).
- 4 Hatton B, Mishchenko L, Davis S, Sandhage KH and Aizenberg J, *Proc Natl Acad Sci USA* **107**:10354–10359 (2010).



- 5 Pursiainen OLJ, Baumberg JJ, Winkler H, Viel B, Spahn P and Ruhl T, *Opt Express* **15**:9552–9561 (2007).
- $6\ \ Baumberg\,JJ, Pursiainen\,OLJ\, and\, Spahn\, P, \textit{Phys Rev\,B}\, \textbf{80}: 201103\ (2009).$
- 7 Snoswell DRE, Kontogeorgos A, Baumberg JJ, Lord TD, Mackley MR, Spahn P et al., Phys Rev E 81:020401 (2010).
- 8 Finlayson CE, Goddard C, Papachristodoulou E, Snoswell DRE, Kontogeorgos A, Spahn P *et al.*, *Opt Express* **19**:3144–3154 (2011).
- 9 Vukusic P and Sambles JR, Nature 424:852-855 (2003).
- 10 Braun PV, Nature 472:423-424 (2011).
- 11 Finlayson CE, Spahn P, Snoswell DRE, Yates G, Kontogeorgos A, Haines Al et al., Adv Mater 23:1540 1544 (2011).
- 12 Snoswell DRE and Baumberg JJ, Textiles 4:8-10 (2009).
- 13 Finlayson CE, Snoswell DRE, Spahn P and Baumberg JJ, Laser Focus World 47:43 (2011).
- 14 Sussman J, Snoswell DRE, Kontogeorgos A, Baumberg JJ and Spahn P, Appl Phys Lett 95:173116 (2009).
- 15 Schäfer CG, Gallei M, Zahn JT, Engelhardt J, Hellmann GP and Rehahn M, Chem Mater 25:2309–2318 (2013).
- 16 Imai Y, Finlayson CE, Goldberg-Oppenheimer P, Zhao Q, Spahn P, Snoswell DRE et al., Soft Matter 8:6280–6290 (2012).

- 17 Zhao Q, Haines AI, Snoswell DRE, Keplinger C, Kaltseis R, Bauer S et al., Appl Phys Lett 100:101902 (2012).
- 18 Yang S-M, Kim S-H, Lim J-M and Yi G-R, *J Mater Chem* **18**:2177 2190 (2008).
- 19 Kontogeorgos A, Snoswell DRE, Finlayson CE, Baumberg JJ, Spahn P and Hellmann GP, Phys Rev Lett 105:233909 (2010).
- 20 Vignolini S, Rudall PJ, Rówland AV, Reed A, Moyroud E, Faden RB et al., Proc Natl Acad Sci USA 109:15712 – 15715 (2012).
- 21 Haines Al, Finlayson CE, Snoswell DRE, Spahn P, Hellmann GP and Baumberg JJ, *Adv Mater* **24**:OP305–OP308 (2012).
- 22 Spahn P, Finlayson CE, Mbi Etah W, Snoswell DRE, Baumberg JJ and Hellmann GP, J Mater Chem 21:8893–8897 (2011).
- 23 Finlayson CE, Haines AI, Snoswell DRE, Kontogeorgos A, Vignolini S, Baumberg JJ et al., Appl Phys Lett 99:261913 (2011).
- 24 Whitney HM, Kolle M, Andrew P, Chittka L, Steiner U and Glover BJ, Science 323:130–133 (2009).